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MOF encapsulated sub-nm Pd skin/Au nanoparticles as antenna-reactor plasmonic catalyst for light driven CO₂ hydrogenation

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ABSTRACT

Activation of CO₂ with plasmon induced hot electrons has attracted great attention due to its moderate reaction conditions, but high-efficiency plasmonic catalysts still remains challenging. Herein, we designedly prepared an antenna-reactor plasmonic catalyst with core-shell structure, i.e. sub-nm Pd skin/Au NPs encapsulated within UiO-66-NH₂ (Au@Pd@UiO-66-NH₂-0.5) for CO₂ hydrogenation under light-heat dual activation. Under the photo-thermal synergism, CO₂ can be efficiently converted into CO and the optimal production rate reaches 3737 μmol/g_{metal}/h at 150 °C. Thereinto, ultrafast transient absorption spectroscopy reveals that the sub-nm Pd skin provides a longer time window for hot electron transfer and efficient retardation of energy dissipation. Theoretical calculations further confirm the significant reductions in HOMO-LUMO gap of CO₂ and reaction energy barrier from CO₂* to COOH* through adsorption of CO₂ on sub-nm Pd skin, promoting its efficiency and selectivity. Our findings provide new insights into design of bimetallic plasmonic catalysts for low-temperature catalysis.

1. Introduction

With growing energy needs and environmental concerns, converting greenhouse gas CO₂ into high-value chemicals and fuels is becoming increasingly important technically and economically due to its great potentials in alleviating the current energy and environmental crisis [1–4]. However, traditional thermal activation of CO₂ usually requires relatively high temperatures and pressures owing to its extremely thermodynamic stability and chemical inertness, thereby leading to huge energy consumption coupled with potential safety hazards and high risk of catalyst deactivation [5–7]. To overcome these issues, photo-assisted catalysis has been used as an effective strategy to reduce the energy barrier of chemical reactions [8–17]. Recent advances have highlighted that hot electrons photoexcited in nanostructures of plasmonic metal (such as Au, Ag, Cu and Al) with strong light absorption can be injected into the frontier orbitals of adsorbed species, thereby reducing the activation energies of reactions under much milder conditions than conventional thermocatalysis [12,18–21]. However, the rational design and synthesis of such plasmonic catalysts with high

efficiency photo-thermal synergistic catalysis are still challenging.

Bimetallic heterostructures (Au-Pt, Al-Pd, Cu-Ru, etc.) consisting of plasmonic metal (antenna) and reactive metal (reactor) nanostructures, known as “antenna-reactor” plasmonic catalysts, have recently attracted considerable interests due to their fantastic optical properties [22,23] and great potential applications in photocatalysis [12,19,22,24]. In such antenna-reactor architectures, the local field induced by photoexcited plasmonic antenna can drive “forced” plasmons to the adjacent catalysis reactor, thereby directly producing hot electrons in the catalytic reactor to drive chemical processes [25]. The heterometallic “antenna-reactor” plasmonic catalysts mainly have focused on hetero-dimer structure, where the plasmonic metal and active metal are independent components [22,24,26]. Notably, the catalysis efficiency of plasmonic catalysts is highly sensitive to the geometric configuration, such as the shape and size of antenna and reactor, as well as the distance between them [26–28]. Structurally, spherical core-shell heterostructures are more favorable for boosting the plasmonic catalysis efficiency of “antenna-reactor” catalysts than their hetero-dimer counterparts owing to the densely connected and strong interaction between them [29–31]. To

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achieve optimal catalysis efficiency, active metal shells serving as reactors should be downsized to the one-nanometer scale for efficient energy transfer from the internal plasmonic antenna to the adsorbed reaction species. However, till now it is still a great challenge to precisely control the architecture of antenna-reactor plasmonic catalysts and clarify key factors behind surface chemical processes.

Herein, novel metal-organic frameworks (MOFs) encapsulated sub-nm (0.5 nm) Pd skin/Au nanoparticles (Au@Pd@UiO-66-NH₂-0.5 NPs) were developed as antenna-reactor plasmonic catalysts for light-driven CO₂ hydrogenation under mild conditions. In this well-designed architecture, the Au NP cores work as an “antenna” to provide an intense localized electric field, while the sub-nm Pd skin play as active sites to activate and hydrogenate CO₂. Our experiments demonstrated that with such sub-nm Pd skin Au NPs being simultaneously encapsulated within amino-modified Zr (IV)-based MOF (UiO-66-NH₂), the highly selective conversion of CO₂ into CO was successfully achieved under illumination. Most strikingly, by controlling the Pd shell at atomic precision, the optimal catalytic activity of such MOF encapsulated sub-nm Pd skin/Au NPs reached 3737 μmol/g_{metal}/h under light-heat dual activation, which is far higher than that of counterparts (Au@UiO-66-NH₂ and Au@Pd@UiO-66-NH₂-2 with 2 nm Pd shell). The influences of the Pd shell thickness and MOF encapsulation structure on the activity and product selectivity of this antenna-reactor plasmonic catalyst were detailedly studied.

2. Experimental section

2.1. Chemicals and materials

Chloroauric acid (HAuCl₄·4H₂O, 99.9%), Pd chloride (II) (PdCl₂, 99.9%), citrate sodium (Na₃C₆H₅O₇·2H₂O, 99%), N, N-dimethylformamide (DMF, 99.5%) and acetic acid (99.5%) were purchased from Sinopharm Chemical Reagent Co., Ltd. Zirconium tetrachloride (ZrCl₄, 98%), polyvinyl pyrrolidone (PVP K30), terephthalic acid (H₂BDC, 99%), and 2-aminoterephthalic acid (H₂ATA, 98%) were purchased from Energy Chemical.

2.2. Material synthesis

2.2.1. Synthesis of Au@Pd NPs

First, 100 mL HAuCl₄·4H₂O (0.1 mg/mL) aqueous solution was poured into a 150 mL round bottom flask and heated to 100 °C under constant stirring. Next, 3 mL citrate sodium (10 g/L) was added to the above HAuCl₄ aqueous solution to yield a wine-red Au colloidal solution after reaction for about 20 min. The resulting solution was then cooled to room temperature followed by the addition of 0.5 g PVP under stirring for 24 h. The Au colloidal suspension was washed several times with DMF at 17,000 rpm for 20 min and re-dispersed in 20 mL DMF. The as-obtained suspension was heated to 100 °C under stirring followed by the addition of 1 mL H₂PdCl₄ aqueous solution (4 mM/L for Au@Pd-0.5 and 10 mM/L for Au@Pd-2). After 2 h reaction, the color of the suspension changed from wine-red to light purple, and the resulting Au@Pd NPs were filtered off and washed several times with DMF at 13,000 rpm for 10 min

2.2.2. Synthesis of Au@Pd@UiO-66-NH₂-0.5

First, 27.3 mg H₂ATA and 100 mg PVP were dissolved in a mixture of 7 mL DMF and 1.2 mL acetic acid. Next, 1 mL Au@Pd-0.5 NPs dispersed in DMF was added to the mixture and stirred under ultrasounds for 5 min. Meanwhile, an amount of 33.4 mg ZrCl₄ was dispersed in 7 mL DMF and then poured into the above mixture and transferred into a 25 mL Teflon-lined stainless-steel autoclave under heating to 120 °C for 24 h. The obtained product was washed several times with DMF and the resulting powder was dried in an oven overnight at 120 °C. Au@UiO-66-NH₂ and Au@Pd@UiO-66-NH₂-2 were obtained under the same conditions by replacing Au@Pd-0.5 with Au NPs and Au@Pd-2 NPs,

respectively.

2.2.3. Synthesis of Au@Pd@UiO-66-NH₂-0.5, Au@Pd/P25-0.5 and Au@Pd/SiO₂-0.5

First, 1 mL Au@Pd-0.5 in DMF was dropwise added to 30 mg pre-synthesized UiO-66-NH₂, commercial P25 or commercial SiO₂, and manually ground for 5 min. The resulting Au@Pd@UiO-66-NH₂, Au@Pd/P25-0.5 and Au@Pd/SiO₂-0.5 was then collected by centrifugation followed by washing several times with DMF and dried overnight in an oven at 120 °C.

2.3. Catalytic tests

2.3.1. Photo-assisted CO₂ hydrogenation measurement of samples

The photo-assisted CO₂ hydrogenation activities of samples were evaluated by vapor-solid reaction mode in a commercial evaluation system (CEL-HPR100T+, Beijing China Education Au-Light Co., Ltd, Fig. S5). To this end, 5 mg catalyst was evenly dispersed on a quartz chip and transferred into a 150 mL stainless steel reactor with a quartz window (diameter: 3.5 cm) and compensatory heating side. After sealing, the reactor was flushed several times to remove the air and then filled with 1 MPa CO₂/H₂ mixed gas (CO₂:H₂ = 1:3). Next, the reactor was irradiated by a 300 W Xe lamp (CEL-HXF300, Beijing China Education Au-Light Co., Ltd.). Meanwhile, the reactor temperature (typically 150 °C) was controlled by a thermocouple and heating system connected to the reactor under both dark and light experiments. The final products were detected by gas chromatography (SHIMADZU GC-2014C) equipped with a flame ionization detector and thermal conductivity detector. UV cut-off filter ($\lambda < 400$ nm) and ($400 < \lambda < 650$ nm) filters were used to explore the influence of light wavelength on the efficiency of catalysts toward the CO₂ hydrogenation.

2.4. Numerical calculation of scattering and heat generation

All the numerical calculations were performed by COMSOL Multiphysics based on finite element methods. The calculation was divided into two steps, where the electromagnetic scattering of the nano-structure was solved first and then the heat generation of the system was calculated. In the first step, a scattered-field formulation was used, where the background field was solved analytically through Fresnel's law in absence of the nanoparticles. The incident power was 900 mW/cm², which was the same as experiments. The diameters of MOF and gold core were 150 nm and 14 nm, and the thickness of palladium shell was 0.5 nm or 2 nm. The permittivity of gold and palladium were taken from experimental data measured by Johnson and Chrwasty [32,33]. In the second step, the heat generation was calculated based on the steady-state heat transfer model. The heat power volume density Q_d , which was contributed by the heat generated in the materials, was written as

$$Q_d = \frac{1}{2} \epsilon_0 \omega \text{Im}(\epsilon_r) |E|^2$$

where ϵ_0 was the permittivity of vacuum; ω was the angular frequency of the light; ϵ_r was the relative permittivity of the material; E was electric field. The thermal diffusion equation in the steady-state regime was written as

$$\nabla \cdot (-k \nabla T) = Q_s$$

where k was the thermal conductivity; T was the temperature; Q_s was the power density of heat source. The thermal conductivity of UiO-66-NH₂ air, gold and palladium were 0.11 W/(m·k), 0.03 W/(m·k), 317 W/(m·k) and 71.8 W/(m·k) respectively.

13.8 nm) were synthesized by the conventional sodium citrate reduction method [36]. A seed-mediated growth was then conducted to form Au@Pd NPs with core-shell structures. In this process, the thickness of Pd shell deposited on Au seed could precisely be controlled from sub-nm (0.5 nm) to 2 nm by adjusting the amount of Pd precursor (H_2PdCl_4). Next, the resulting Au@Pd NPs were encapsulated within UiO-66-NH₂ with strong adsorption ability toward CO₂ to protect Au@Pd NPs from aggregation. To observe the structure evolution of products, transmission electron microscopy (TEM) characterization was detailedly conducted on the Au@Pd NPs with 0.5 nm Pd shell before and after encapsulation (i.e., Au@Pd-0.5 and Au@Pd@UiO-66-NH₂-0.5).

Compared to the initial Au seeds, the diameter of Au@Pd-0.5 NPs increases to 14.6 ± 1.3 nm (Fig. S1). The spherical aberration (Cs) corrected high-angle annular dark-field scanning transmission electron microscopy (HAADF-STEM) image and corresponding elemental mappings show the surface of Au NPs is covered by a thin layer of Pd (Fig. 1b–e). To determine the accurate thickness of Pd shell, atomic-resolution Cs-corrected HAADF-STEM images were recorded and the results are displayed in Fig. 1f–g. The whole Au@Pd-0.5 particle exhibits continuous atom arrangements corresponding to (111) planes. However, a careful survey reveals slightly smaller interatomic distances of the two most superficial atomic layers of NP when compared to the

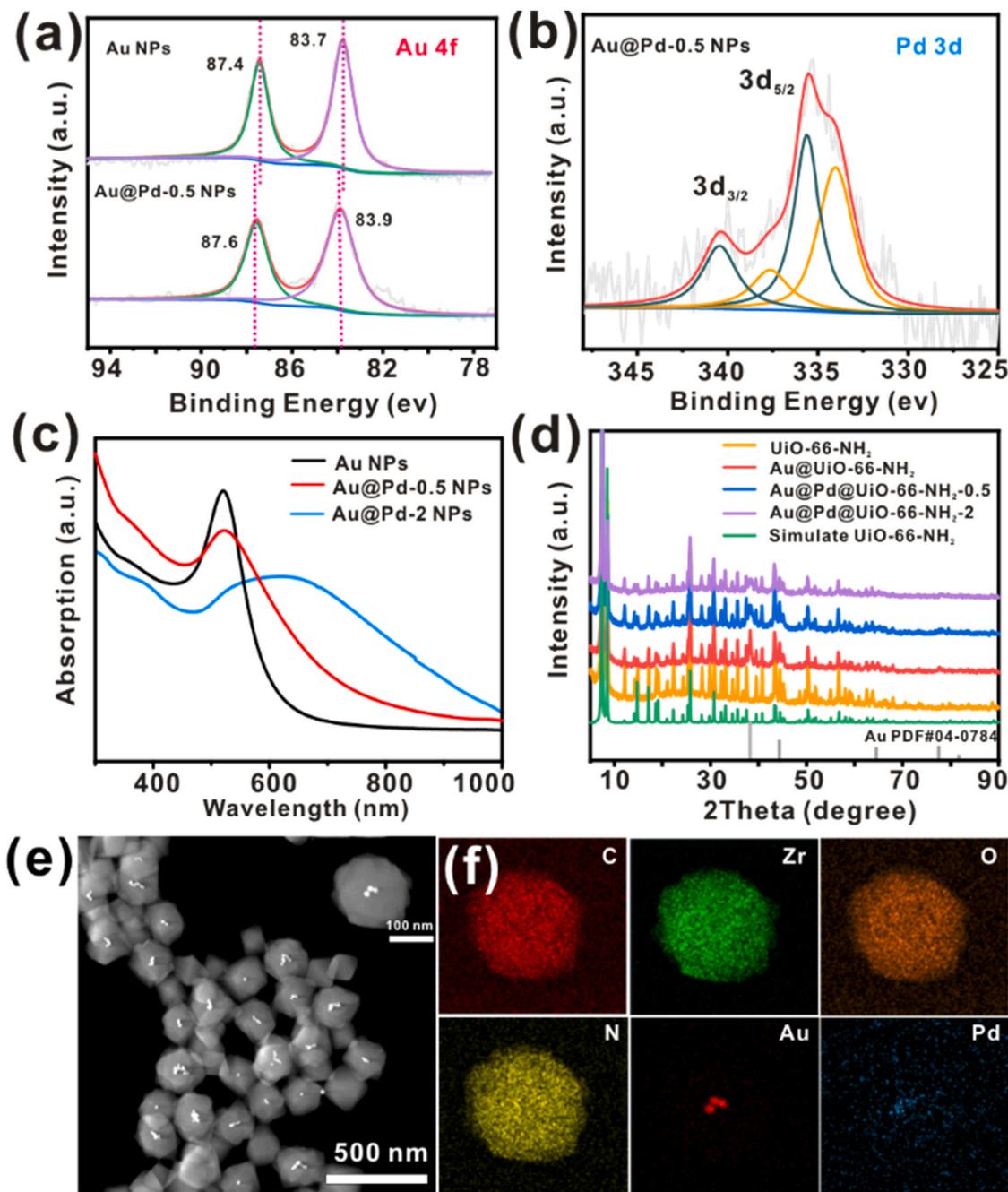


Fig. 2. (a) Au 4f high-resolution XPS spectra of Au NPs and Au@Pd-0.5 NPs. (b) Pd 3d high-resolution XPS spectra of Au@Pd-0.5 NPs. (c) UV–vis spectra of Au NPs, Au@Pd-0.5 NPs, and Au@Pd-2 NPs. (d) Powder XRD patterns of UiO-66-NH₂, Au@UiO-66-NH₂, Au@Pd@UiO-66-NH₂-0.5, Au@Pd@UiO-66-NH₂-2, and simulate UiO-66-NH₂. (e) HAADF-STEM image of Au@Pd@UiO-66-NH₂-0.5 catalysts. The inset shows a magnified image of single Au@Pd@UiO-66-NH₂-0.5. (f) Corresponding EDS maps of single Au@Pd@UiO-66-NH₂-0.5 in (e).

inner region, which should be caused by the smaller atomic diameter of Pd with respect to that of Au (Fig. 1h–i). These results reveal that the shell is only composed of two layers of Pd atoms with a thickness of ca. 0.5 nm, consistent with the increase in the Au@Pd-0.5 NPs size.

The structural homogeneity of such sub-nm Pd skin/Au NPs was further confirmed by cyclic voltammetry measurements. In the first cycle, only a well-defined Pd oxide reduction peak emerges at 0.78 V (Fig. S2), indicating Au NPs are fully covered by ultrathin Pd shells. The Pd shell lattices slightly expand due to the effect of Au core, which influences the electronic structures of Au@Pd NPs [37]. X-ray photoelectron spectroscopy (XPS) analyses showed that Au 4f peaks of Au@Pd-0.5 NPs shift to higher binding energies when compared to pristine Au NPs, suggesting electron transfer from Au to Pd (Fig. 2a and b). Additionally, after the coverage of Au NPs by Pd atoms, the LSPR absorption peak of Au at 520 nm becomes significantly broad (Fig. 2c) [38,39]. All these results indicate that the electronic structure of Au@Pd-0.5 NPs is different from that of Au NPs due to the introduction of such sub-nm Pd shell.

The resulting sub-nm Pd skin/Au NPs were then encapsulated in UiO-66-NH₂ by a simple solvothermal route using 2-aminoterephthalic acid (H₂ATA, 98%) and ZrCl₄ as sources and DMF as solvent at 120 °C for 24 h. Powder X-ray diffraction (PXRD) reveals that all the diffraction peaks of Au@Pd@UiO-66-NH₂-0.5 match well with the simulated pattern of UiO-66-NH₂ except for the weak peak at 38° related to Au (111) plane (Fig. 2d). Meanwhile, the diffraction peaks of Pd are absent due to its relatively lower content (0.17 wt%) compared to Au (1.20 wt%) in Au@Pd@UiO-66-NH₂-0.5. And the contents of Au and Pd in Au@Pd@UiO-66-NH₂-2 is 1.32 wt% and 0.36 wt% obtained by ICP-MS. The SEM images of UiO-66-NH₂ are shown in Fig. S3. After the encapsulation of Au@Pd-0.5 NPs in UiO-66-NH₂, the whole particles display irregular polyhedral shapes with diameters of 150–200 nm (Fig. S4). The HAADF-STEM images and corresponding element mapping images (Fig. 2e and f) reveal more than one Au@Pd NPs are encapsulated in the central region in most Au@Pd@UiO-66-NH₂-0.5 particles. Such status of

one-to-many encapsulation of Au@Pd NPs within UiO-66-NH₂ will enhance the catalytic performance of plasmonic catalysts since the plasmonic coupling induced by interparticle interactions can further intensify the electromagnetic energy in the nearfield centered on the nanostructures [40].

The catalytic properties of as-obtained Au@Pd@UiO-66-NH₂-0.5 toward CO₂ hydrogenation were evaluated in the gas-solid phase of a 150 mL batch reactor. (Fig. S5). Two control catalysts (Au@UiO-66-NH₂ and Au@Pd@UiO-66-NH₂-2) were prepared to investigate the role of Pd shell, which were respectively encapsulated with the pristine Au NPs (Fig. S6) and Au@Pd NPs with 2 nm shell (Fig. S7). Compared to Au@UiO-66-NH₂, the Au LSPR peak of Au@Pd@UiO-66-NH₂ at 520 nm significantly weakens or vanishes as the thickness of Pd shell increases (Fig. S8) [41]. Furthermore, the Zr 3d high-resolution XPS spectra of Au@Pd@UiO-66-NH₂-0.5 and Au@UiO-66-NH₂ significantly shift to lower binding energies when compared to UiO-66-NH₂, indicating strong interactions between metals and UiO-66-NH₂ (Fig. S9) [42].

The catalytic activities of Au@Pd@UiO-66-NH₂-0.5 toward the CO₂ hydrogenation reaction are gathered in Fig. 3a. Without illumination, the CO₂ molecules cannot be activated at room temperature, and no products are detected even at an elevated temperature of 150 °C. By contrast, CO₂ is efficiently transformed into CO over Au@Pd@UiO-66-NH₂-0.5 under irradiation of 300 W Xe lamp without additional heating, with the CO production rate reaching 758 μmol/g_{metal}/h. As the reaction temperature rises, the CO production rate significantly increases and reaches a maximum of 3737 μmol/g_{metal}/h at 150 °C, which is 5-fold higher than that obtained solely by light without heat (Fig. 3b). Additionally, the photocatalytic rate using Au@Pd@UiO-66-NH₂-0.5 catalyst varies linearly with the light intensity (Fig. S10), indicating that the catalytic reaction over Au@Pd@UiO-66-NH₂-0.5 follows the hot electrons driven mechanism [9,10]. Also, the CO production rate over Au@Pd@UiO-66-NH₂-0.5 varies linearly with the reaction time (Fig. S11). The TEM and XRD analyses of recycled catalysts demonstrate that the morphology and structure of Au@Pd@UiO-66-NH₂-0.5 is well

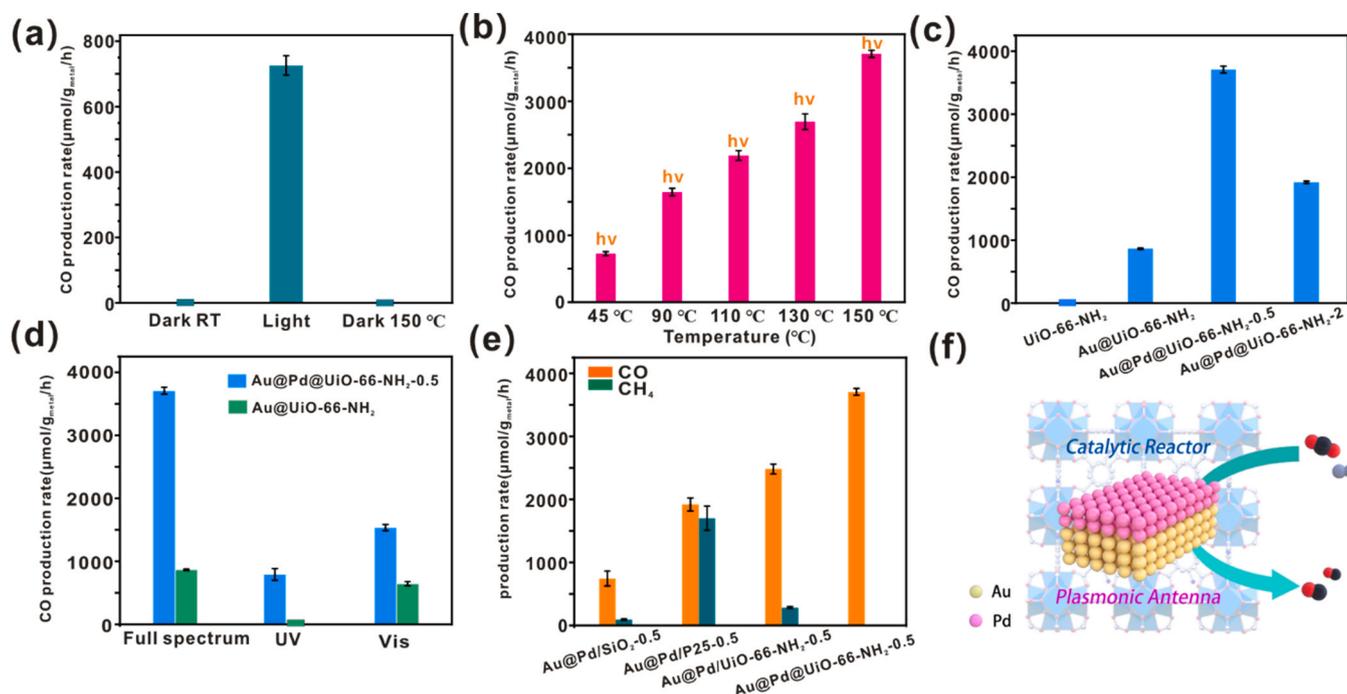


Fig. 3. (a) CO production rates using Au@Pd@UiO-66-NH₂-0.5 under dark at room temperature, under irradiation with a 300 W Xe lamp, and under dark at 150 °C. (b) CO production rates using Au@Pd@UiO-66-NH₂-0.5 at different temperatures under illumination. Production rates obtained by the prepared catalysts consisting of different: (c) metallic catalysts, (d) wavelengths, and (e) loading styles at 150 °C under illumination. (f) Scheme of the core-shell structure of Au@Pd@UiO-66-NH₂-0.5.

maintained after reaction under illumination at 150 °C for 6 h (Fig. S12). It is worth noting that the CO production rate per unit time started to decline slightly when reaction time was further extended to 8 h. To confirm the performance stability of Au@Pd@UiO-66-NH₂-0.5, cycling test were conducted. The cycling test shows that the catalytic performance dropped by 24% at second cycle and maintained by 62% at third cycle (Fig. S13). TEM images of the sample after the cycling testing suggested that nanoparticles underwent sintering and the part of the MOF structure is broken (Fig. S14).

The catalytic activities of CO₂ hydrogenation of Au@UiO-66-NH₂, Au@Pd@UiO-66-NH₂-2, and the blank UiO-66-NH₂ were measured under the same conditions. As shown in Fig. 3c, no product is detected over UiO-66-NH₂. By comparison, a CO production rate of 876 μmol/g_{metal}/h is achieved over Au@UiO-66-NH₂, less than 1/4 of that recorded over Au@Pd@UiO-66-NH₂-0.5. However, the catalytic activity of Au@Pd@UiO-66-NH₂ catalyst (Au@Pd@UiO-66-NH₂-2) decreases sharply to half (1911 μmol/g_{metal}/h) of that of Au@Pd@UiO-66-NH₂-0.5 when the thickness of the Pd shell increases to 2 nm. Undoubtedly, Pd serving as a ‘reactor’ material is more active toward CO₂ hydrogenation than Au, and the enhancement effect of Pd shell strongly depends on its thickness. Besides, the sub-nm Pd shell exhibits distinct enhancement effects under different wavelengths when compared to Au cores (Fig. 3d). In particular, the activity of Au@Pd@UiO-66-NH₂-0.5 in the visible light region (400 < λ < 650 nm) is twice that obtained in the ultraviolet region (λ < 400 nm). It is possible that there is more energy transfer to the Pd shell in this wavelength range (400 < λ < 650 nm) due to the difference in dielectric function of Au to Pd [31].

To explore the role of the MOF encapsulated structure, different carriers and loading status of Au@Pd NPs are further tested (Figs. S15 and S16). For Au@Pd-0.5 NPs supported on inert SiO₂ carrier, CO₂ can be converted into CO accompanied by small amounts of CH₄ and the CO production rate is only 788 μmol/g_{metal}/h (Fig. 3e and Table S1). When Au@Pd-0.5 NPs supported on TiO₂, CO₂ is converted into CO and CH₄,

with the production rate reaching 1920 μmol/g_{metal}/h and 1702 μmol/g_{metal}/h. (Fig. 3e and Table S1). The poor CO selectivity may be related to the form of CO₂ adsorbed on TiO₂. By comparison, Au@Pd@UiO-66-NH₂-0.5 built with Au@Pd-0.5 NPs directly loaded on MOF shows enhanced catalytic activity reaching 2494 μmol/g_{metal}/h. However, the activity of Au@Pd@UiO-66-NH₂-0.5 greatly reduces and small amounts of CH₄ are produced in addition to CO when compared to Au@Pd@UiO-66-NH₂-0.5 with an encapsulated structure. Thus, the encapsulation process enables UiO-66-NH₂ shell with better CO₂ enrichment effect and stronger interaction with active metal catalysts, greatly improving the catalytic activity [43,44]. Additionally, the MOF shell have the ability to control the adsorption states of reactant/product molecules on the catalyst surface, thereby improving the product selectivity [45–49]. To the best of our knowledge, the Au@Pd@UiO-66-NH₂-0.5 also exhibited an excellent activity performance compared with other representative catalysts recently reported (Table S2). A schematic diagram of light-driven CO₂ hydrogenation over Au@Pd@UiO-66-NH₂-0.5 is displayed in Fig. 3f.

The above data reveal the importance of both the thickness of Pd shell and MOF encapsulation structure in controlling the activity and product selectivity of such special antenna-reactor plasmonic catalysts. Hence, theoretical simulation and ultrafast transient absorption (TA) spectroscopy were conducted to clarify the enhancement mechanism. As we know, plasmonic metallic nanostructures have the ability to convert light to thermal energy via electron–phonon coupling, thus providing localized heating (i.e., the photothermal effect), which is considered to be another possible reason for improving the performances of plasmonic hybrid photocatalysts [50]. In general, only strong light intensity will create an obvious photothermal effect [10]. To this end, The steady temperatures of Au and Au@Pd NPs encapsulated within UiO-66-NH₂ at 520 nm illumination were first simulated by COMSOL Multiphysics based on finite element methods to confirm the photothermal effect on the catalytic performances [51]. The increased steady-state

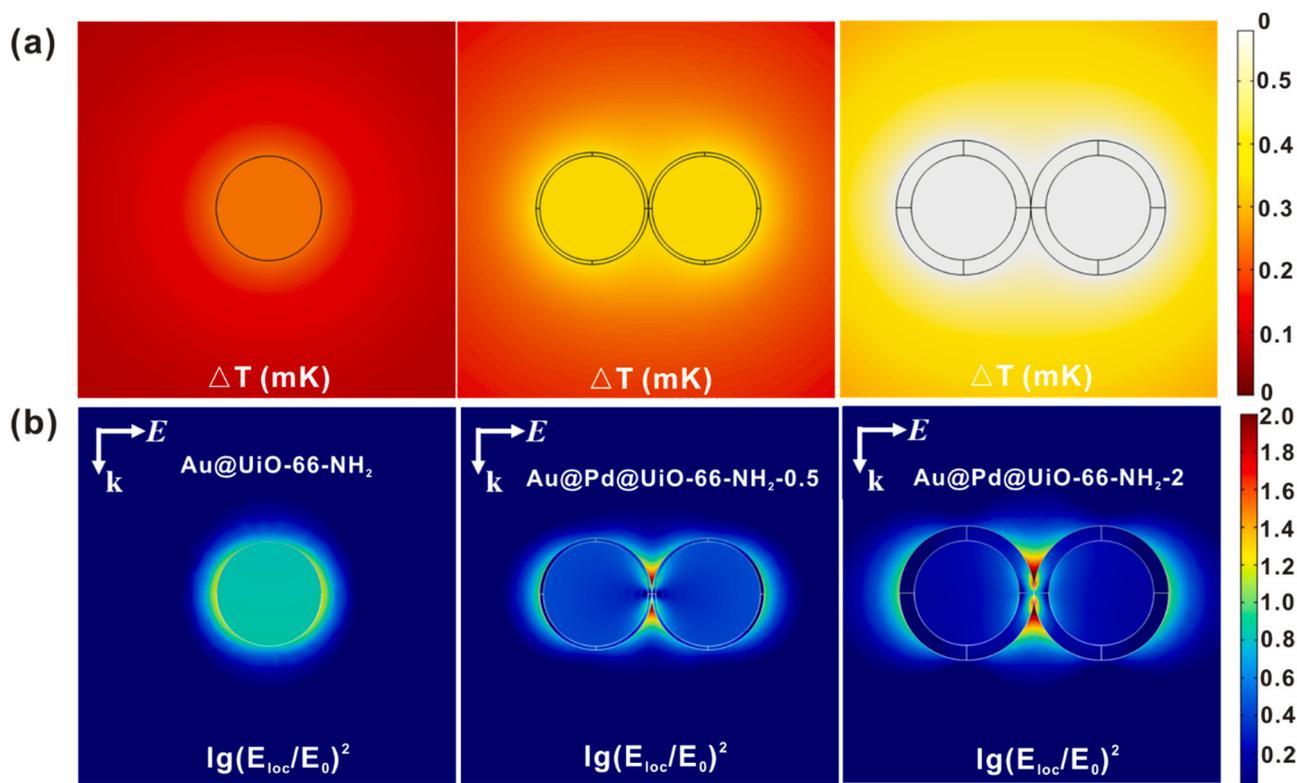


Fig. 4. (a) Increase in steady-state temperature caused by 520 nm light excitation at an incident power density of 0.9 W/cm². (b) Enhanced electric field distributions of Au@UiO-66-NH₂, Au@Pd@UiO-66-NH₂-0.5, and Au@Pd@UiO-66-NH₂-2. E_{loc} represents the local electric field and E_0 is the incident electric field.

temperatures of Au@UiO-66-NH₂, Au@Pd@UiO-66-NH₂-0.5 and Au@Pd@UiO-66-NH₂-2 are recorded as 0.24, 0.36 and 0.58 mK, respectively (Fig. 4a). Thus, the results indicated that there were differences in relative heating between two types of nanoparticles. On the other hand, the electric field enhancement distributions of related samples were also simulated. As shown in Fig. 4b, the Au core acts as an "antenna" to generate strong localized electromagnetic fields, as well as enhances the near field of the Pd shell. For Au@Pd@UiO-66-NH₂, multiple Au@Pd NPs are encapsulated together within UiO-66-NH₂ to generate hot spots between them and accelerate the production of hot electrons [40]. However, the light absorption of the Au@Pd NPs with thick Pd shell significantly weakens (Figs. S8 and S17). This result partly accounts for the lower activity of Au@Pd@UiO-66-NH₂-2 compared to Au@Pd@UiO-66-NH₂-0.5.

To further explore the plasmon-enhanced reaction mechanism, ultrafast TA measurements were carried out to track the carrier dynamics in plasmonic Au, Au@Pd-0.5, and Au@Pd-2. In Fig. 5a, the negative bleach signals around 520 nm with the two positive wings have been attributed to the broadening of plasmon band [52]. The decay of these features reflects the carrier and lattice cooling. The hot carriers quickly dissipate their excess energy to the lattices via electron-phonon (e-p) coupling, which is followed by a slower heat transfer from the lattice to environment via phonon-phonon scattering (p-p). Thus, the kinetics of the plasmon band bleach can be described by a two-step cooling model [52,53]. As shown in Fig. 5b, the kinetics fitting reveals that the e-p processes in Au@Pd-0.5 ($\tau_1 = 0.7$ ps) and Au@Pd-2 ($\tau_1 = 0.6$ ps) are faster than that in Au ($\tau_1 = 1.6$ ps). This phenomenon is likely due to the larger electron-phonon coupling constant in Pd ($8.7 \times 10^{17} \text{ W m}^{-3} \text{ K}^{-1}$) when compared to that of Au ($3.0 \times 10^{16} \text{ W m}^{-3} \text{ K}^{-1}$) [54]. In contrast to Au@Pd-2, Au@Pd-0.5 with sub-nm Pd shell exhibits a longer hot-electron decay time. As a result, the presence of sub-nm Pd skin could provide a longer time window for additional hot electron transfer process. Meanwhile, the p-p rate also declines as thickness of Pd increases, suggesting efficient retardation of energy dissipation caused by the Pd shell [54,55]. In other words, the Pd shell helps to retain the

energy gained from photon absorption within the NPs for a longer duration. Therefore, it is necessary to precisely control the thickness of Pd shell on plasmonic Au core in order to obtain the optimal catalytic activity.

In general, the electronic structure of CO₂ molecules would change after adsorption on metal surfaces [10,56,57]. To explore the influence of Pd shell on the surface chemical reactions, the electronic structures were calculated for free CO₂ adsorbed on Au sites in Au@UiO-66-NH₂, as well as free CO₂ adsorbed on Pd and Zr sites at the interface of Au@Pd-0.5 and UiO-66-NH₂ in Au@Pd@UiO-66-NH₂-0.5. As shown in Fig. 6a, the LUMO (2π antibonding) of CO₂ decreases significantly after the adsorption of CO₂ on the metal surface. The gap between HOMO (5σ bonding) and LUMO ($2\pi^*$) reduces from 8.5 eV for free CO₂ molecules to 3.1 eV for CO₂ adsorbed on Pd surface. This change is much lower than those of CO₂ adsorbed on Au (4.6 eV) and Zr sites at the interface between Au@Pd-0.5 and MOF (3.5 eV). The reason for this has to do with the adsorption of CO₂ on Pd shell, which facilitates the jump of hot-electrons into LUMO of CO₂ under light irradiation due to the significantly reduced gap [10].

First-principles calculations based on spin-polarized density functional theory (DFT) and generalized gradient approximation (GGA) were further used to study the adsorption of CO₂ molecules at different sites of the catalysts. The differential charge densities of CO₂ at Au sites of Au@UiO-66-NH₂, Pd, and Zr sites in Au@Pd@UiO-66-NH₂-0.5 are shown in Fig. 6b. The charge transfer values between CO₂ and each site are estimated to 0.17 eV at Au, 0.31 eV at Pd, and 0.23 eV at Zr. The electron transfer between CO₂ and Pd sites looks more obvious than others. Besides, the free energies of different reaction paths over Au@UiO-66-NH₂ and Au@Pd@UiO-66-NH₂-0.5 are calculated to rationalize the activity and selectivity of catalysts for photothermal CO hydrogenation (Figs. S18–S21). The rate-determining step of the reaction is determined as the formation of COOH* from CO₂* [58]. As shown in Fig. 6c, the sub-nm Pd shell effectively reduces the energy barrier for the generation of COOH* from 1.867 eV to 0.878 eV. According to the free energies of producing CO and CH₄, the reaction is more inclined to

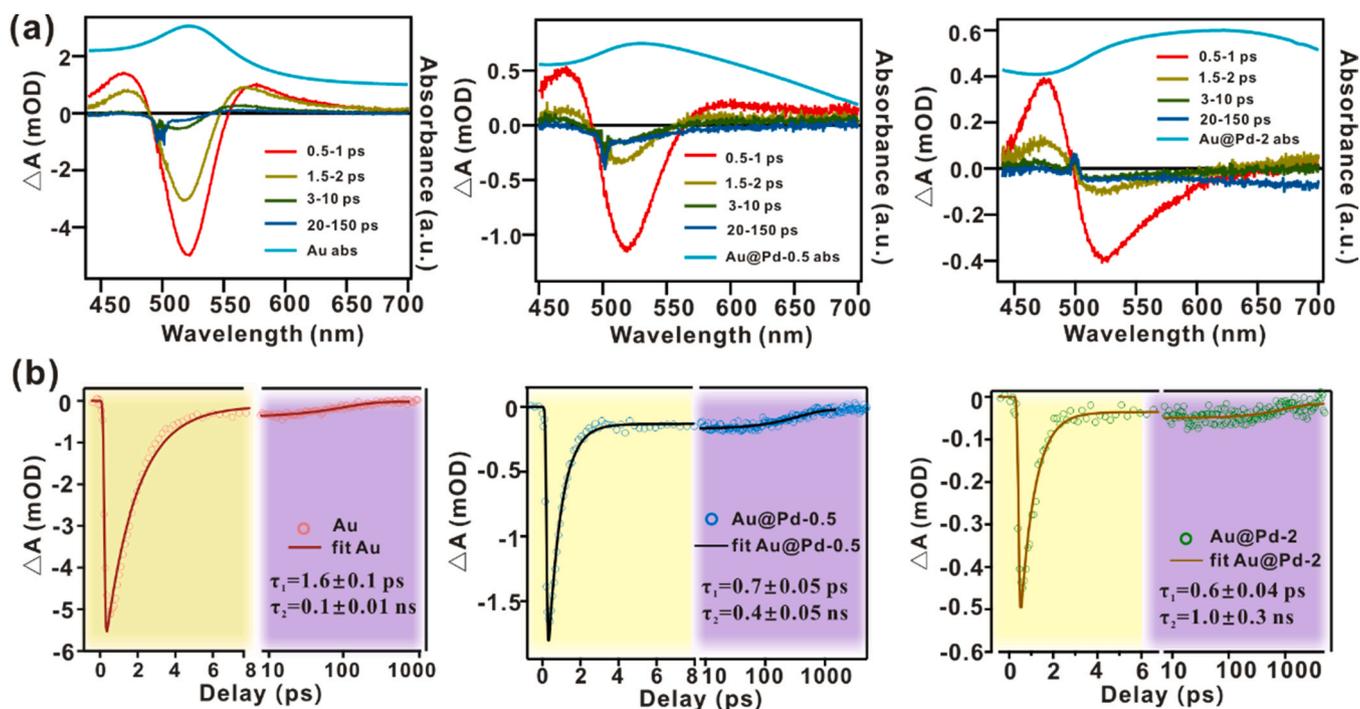


Fig. 5. (a) Ultrafast TA spectroscopy of Au, Au@Pd-0.5, and Au@Pd-2. (b) TA signals from kinetic curves of Au, Au@Pd-0.5, and Au@Pd-2 (τ_1 : electron-phonon scattering and τ_2 : phonon-phonon scattering).

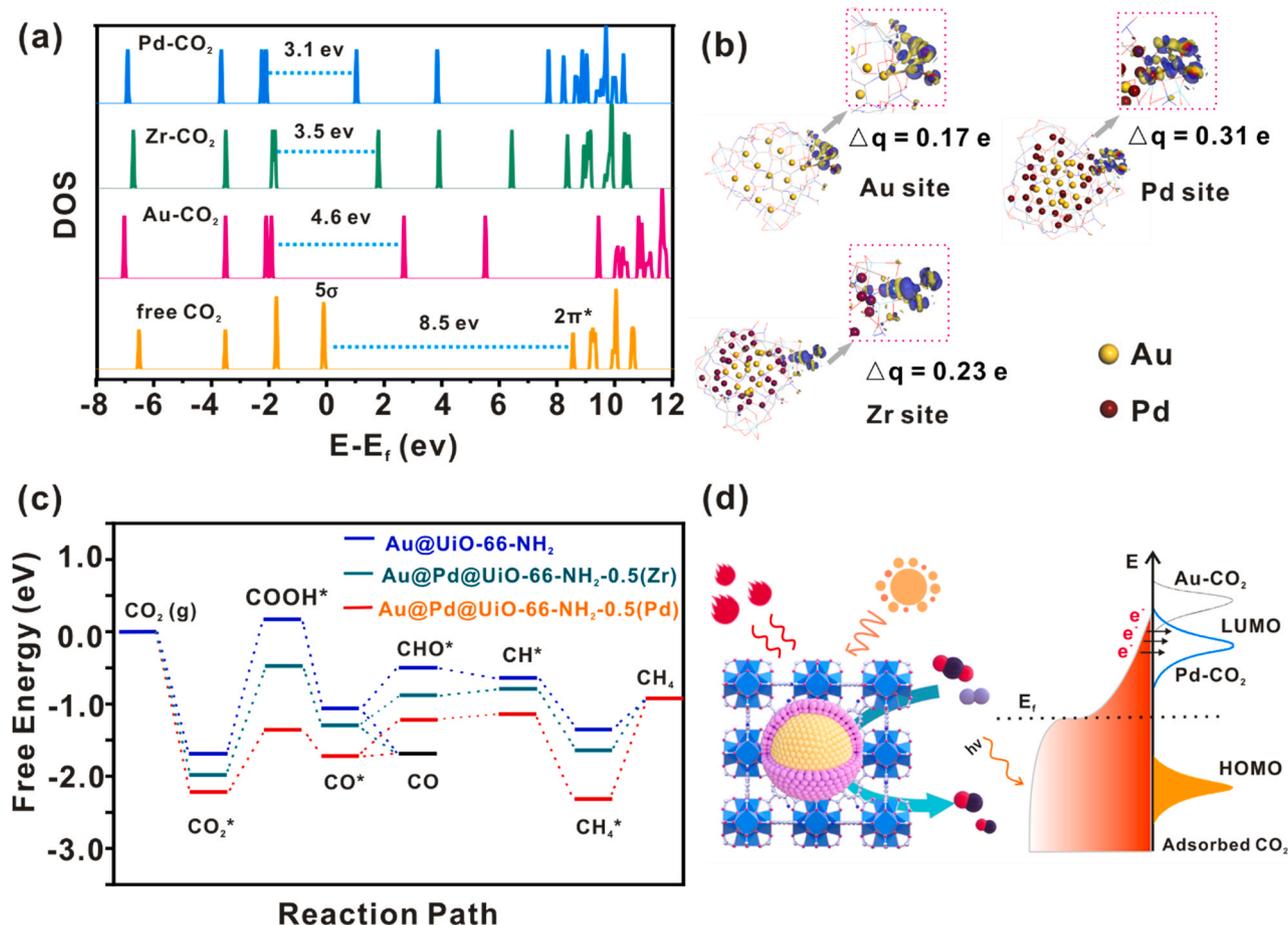


Fig. 6. (a) The electronic structures of free CO₂, as well as CO₂ adsorbed on Au sites in Au@UiO-66-NH₂, Pd, and Zr sites in Au@Pd@UiO-66-NH₂-0.5. (b) Differential charge densities of CO₂ at different sites. The isosurface value is 0.03 eV/Å³. (c) Free energies of different reaction paths over different catalysts and active sites. (d) Schematics of light driven CO₂ hydrogenation over Au@Pd@UiO-66-NH₂-0.5.

produce CO, which is in line with our experimental results. It is worth noting that a comparison between the free energies of the two different reaction paths suggests that CO₂ is more favorable to activate on the Pd surface [59]. In short, the adsorption of CO₂ on sub-nm Pd skin significantly reduces the HOMO-LUMO gap of CO₂, as well as the reaction energy barrier from CO₂* to COOH*. This, in turn, promotes the hot electrons for injection into the LUMO orbitals of CO₂, leading to efficient surface reactions (Fig. 6d).

4. Summary

A novel plasmonic catalyst made of sub-nm Pd skin/Au NPs encapsulated within UiO-66-NH₂ was successfully prepared for the CO₂ hydrogenation reaction under light-heat dual activation. The hot electrons excite by light effectively activated the CO₂ molecules under Xe lamp irradiation and the CO production rate reaches 3737 μmol/g_{metal}/h at 150 °C. The ultrathin Pd shell of 0.5 nm in Au@Pd@UiO-66-NH₂-0.5 significantly improves the catalytic activity. Besides, the encapsulation structure of UiO-66-NH₂ promotes the conversion and selectivity to CO due to the elevated adsorption capability towards CO₂ and strong confinement effect of MOFs. The ultrafast TA spectroscopy data revealed that the thickness of Pd shell greatly affected the decay process of the plasmon. The theoretical calculations demonstrated the effective reduction in both HOMO-LUMO gap and reaction energy barrier of CO₂* to COOH* by sub-nm Pd skin. We believe that the findings presented in this work is of great significance for the future design of plasmon-

enhanced photocatalysts, as well as a better understanding of plasmonic catalysis mechanisms.

CRediT authorship contribution statement

Xibo Zhang: Conceptualization, Methodology, Investigation, Writing - original draft preparation, Writing - review & editing. **Yunyan Fan:** Investigation, Writing - original draft preparation. **Enming You:** Formal analysis. **Zexuan Li:** Investigation. **Yongdi Dong:** Investigation. **Luning Chen:** Investigation. **Ye Yang:** Formal analysis. **Zhaoxiong Xie:** Supervision. **Qin Kuang:** Writing - original draft preparation, Writing - review & editing, Project administration, Funding acquisition. **Lansun Zheng:** Supervision.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

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Appendix A. Supporting information

Supplementary data associated with this article can be found in the online version at [doi:10.1016/j.nanoen.2021.105950](https://doi.org/10.1016/j.nanoen.2021.105950).

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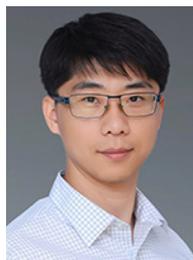
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